

β -Ga₂O₃ on insulator field-effect transistors with drain currents exceeding 1.5 A/mm and their self-heating effect

Hong Zhou, Kerry Maize, Gang Qiu, Ali Shakouri, and Peide D. Ye

Citation: *Appl. Phys. Lett.* **111**, 092102 (2017); doi: 10.1063/1.5000735

View online: <http://dx.doi.org/10.1063/1.5000735>

View Table of Contents: <http://aip.scitation.org/toc/apl/111/9>

Published by the [American Institute of Physics](#)

Articles you may be interested in

[Modulation-doped \$\beta\$ -\(Al_{0.2}Ga_{0.8}\)₂O₃/Ga₂O₃ field-effect transistor](#)

Applied Physics Letters **111**, 023502 (2017); 10.1063/1.4993569

[Electron paramagnetic resonance study of neutral Mg acceptors in \$\beta\$ -Ga₂O₃ crystals](#)

Applied Physics Letters **111**, 072102 (2017); 10.1063/1.4990454

[Terahertz spectroscopy of an electron-hole bilayer system in AlN/GaN/AlN quantum wells](#)

Applied Physics Letters **111**, 073102 (2017); 10.1063/1.4996925

[High responsivity in molecular beam epitaxy grown \$\beta\$ -Ga₂O₃ metal semiconductor metal solar blind deep-UV photodetector](#)

Applied Physics Letters **110**, 221107 (2017); 10.1063/1.4984904

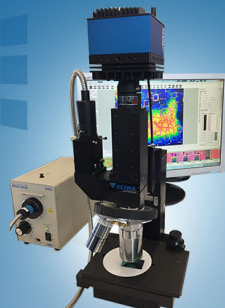
[Improved interface properties of GaN-based metal-oxide-semiconductor devices with thin Ga-oxide interlayers](#)

Applied Physics Letters **110**, 261603 (2017); 10.1063/1.4990689

[Highly conductive homoepitaxial Si-doped Ga₂O₃ films on \(010\) \$\beta\$ -Ga₂O₃ by pulsed laser deposition](#)

Applied Physics Letters **111**, 012103 (2017); 10.1063/1.4991363

The logo for SEIWA OPTICAL features the word "SEIWA" in a bold, white, sans-serif font with a stylized white graphic element to its left consisting of three horizontal bars of varying lengths. Below "SEIWA" is the word "OPTICAL" in a smaller, white, sans-serif font.



NEW IR-2200 Microscope

For fast performance and high precision measurements

[LEARN MORE](#) 

β -Ga₂O₃ on insulator field-effect transistors with drain currents exceeding 1.5 A/mm and their self-heating effect

Hong Zhou, Kerry Maize, Gang Qiu, Ali Shakouri, and Peide D. Ye^{a)}

School of Electrical and Computer Engineering and Birck Nanotechnology Center, Purdue University, West Lafayette, Indiana 47907, USA

(Received 12 April 2017; accepted 17 August 2017; published online 28 August 2017)

We have demonstrated that depletion/enhancement-mode β -Ga₂O₃ on insulator field-effect transistors can achieve a record high drain current density of 1.5/1.0 A/mm by utilizing a highly doped β -Ga₂O₃ nano-membrane as the channel. β -Ga₂O₃ on insulator field-effect transistor (GOOI FET) shows a high on/off ratio of 10¹⁰ and low subthreshold slope of 150 mV/dec even with 300 nm thick SiO₂. The enhancement-mode GOOI FET is achieved through surface depletion. An ultra-fast, high resolution thermo-reflectance imaging technique is applied to study the self-heating effect by directly measuring the local surface temperature. High drain current, low R_c, and wide bandgap make the β -Ga₂O₃ on insulator field-effect transistor a promising candidate for future power electronics applications. *Published by AIP Publishing.* [<http://dx.doi.org/10.1063/1.5000735>]

Recently, β -Ga₂O₃ has shown its great promise for next generation high power device applications due to its ultra-wide bandgap of 4.6–4.9 eV.¹ This allows β -Ga₂O₃ to possess a corresponding empirical estimated electrical breakdown field (E_{br}) of 8 MV/cm, which is several times higher than GaN and SiC. Despite the very early development stage, depletion-mode (D-mode) β -Ga₂O₃ metal-oxide-semiconductor field-effect transistors (MOSFETs) have demonstrated a high blocking voltage of 750 V and an E_{br} of 3.8 MV/cm (Refs. 2 and 3) and enhancement-mode (E-mode) MOSFET has shown a breakdown voltage (BV) of more than 600 V.⁴ β -Ga₂O₃ metal-semiconductor field-effect transistors (MESFETs) and Schottky barrier diodes (SBDs) have also demonstrated a BV of 257 V and 1000 V, respectively.^{5,6} In addition to the excellent direct current (DC) breakdown characteristic, a high pulsed drain current (I_D) of 478 mA/mm has also been achieved for β -Ga₂O₃ MOSFETs.⁷ Until very recently, radio frequency (RF) performance of β -Ga₂O₃ MOSFET was demonstrated, yielding cut off frequency and maximum oscillation frequency (f_T and f_{max}) of 3.3 GHz and 12.9 GHz, respectively.⁸ Meanwhile, β -Ga₂O₃ also has the advantage of a low-cost native bulk substrate that can be synthesized in large size through melt-grown Czochralski, edge defined film fed growth, and floating zone method.^{9–14} However, a crucial disadvantage of this material is its low thermal conductivity of 0.1–0.3 W/cm·K depends on its various crystal orientation.^{15,16} One of the approaches to solve the low thermal conductivity issue of β -Ga₂O₃ is to introduce a high thermal conductivity substrate rather than the β -Ga₂O₃ native substrate. Our previous study has demonstrated a high performance β -Ga₂O₃ on insulator field-effect transistors (GOOI FETs) by transferring β -Ga₂O₃ nano-membrane or nano-belts to SiO₂/Si substrate with SiO₂ thickness of 300 nm to mitigate gate-drain overlap breakdown and serve as gate dielectric as well.¹⁷ The monoclinic structure of bulk β -Ga₂O₃ crystals would allow a facile cleavage into nano-membrane along the [100] direction even though β -Ga₂O₃ is not a van der Waals

2D material, possibly due to its large lattice constant of 12.23 Å along this [100] direction.¹⁸

To realize all merits of β -Ga₂O₃ as a power device, there are still some challenges to be encountered ahead. The reported D/E mode maximum drain current density (I_{DMAX}) for GOOI FETs are 600/450 mA/mm and continuous wave DC of 200 and 2 mA/mm for β -Ga₂O₃ homoepitaxial D/E mode MOSFETs, respectively.^{19,20} How to further enhance the device performance in terms of higher I_{DMAX} and lower on-resistance (R_{on}) and make them comparable to GaN and SiC technologies remained to be demonstrated. In this letter, we have shown that by increasing the doping concentration of the β -Ga₂O₃ nano-membrane and further scale the device, the I_{DMAX} of D/E-mode GOOI FETs can achieve a record value of 1.5/1.0 A/mm, which is around twice of previous record I_{DMAX}. Our results reveal that the contact resistance (R_c) can also be reduced by highly doped channel. Finally, we have evaluated the thermal effect of GOOI FETs and observed the pronounced self-heating effect by using an ultra-fast, high resolution thermo-reflectance (TR) imaging technique.

In our experiment, n-type Sn doping concentration (n) of β -Ga₂O₃ is determined to be $8.0 \times 10^{18} \text{ cm}^{-3}$ from Capacitance-Voltage (C-V) measurement.²¹ The thickness of (100) β -Ga₂O₃ nano-membranes vary from 50 to 70 nm determined by atomic force microscopy (AFM) measurements with surface roughness of 0.33 nm, and the D/E-mode devices are achieved based on the thickness of the nano-membranes. More details about the device fabrication can be found in our previous work.¹⁷ For comparison, devices with lower doping concentration of $3.0 \times 10^{18} \text{ cm}^{-3}$ were also fabricated. Figures 1(a) and 1(b) are device schematic and AFM image of a fabricated device, respectively. The device electric characterizations were carried out with Keithley 4200 Semiconductor Parameter Analyzer. A Microsanj TR system with a high-speed light-emitting diode (LED) pulse and a synchronized charge coupled device (CCD) camera was used for the thermal measurement.^{22–25}

Figure 2(a) presents the DC output characteristics (I_D-V_{DS}) of D-mode GOOI FETs. Devices have a channel

^{a)}Author to whom correspondence should be addressed: yep@purdue.edu

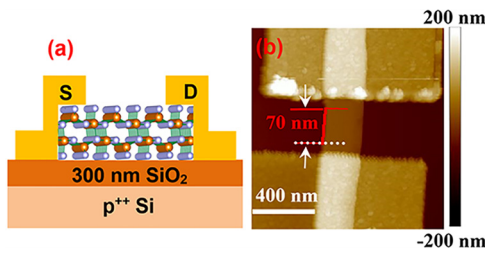


FIG. 1. (a) Schematic cross-section view of a GOOI FET with a 300 nm SiO₂ layer on Si substrate and (b) AFM image of fabricated GOOI FET with β -Ga₂O₃ nano-membrane thickness of 70 nm.

length (L_{CH} , also source to drain spacing L_{SD}) of 0.3 μ m, channel width (W) of 0.15 μ m and channel thickness (t) of 70 nm for $8.0 \times 10^{18} \text{ cm}^{-3}$, and $W = 0.6 \mu\text{m}$ and $t = 100 \text{ nm}$ for $3.0 \times 10^{18} \text{ cm}^{-3}$ device. The device dimensions are accurately determined by scanning electron microscopy (SEM). A record high $I_{D\text{MAX}}$ of 1.5 A/mm is obtained, which is more than 2 times of lower doping channel. Compared with our previous work with $L_{CH} = 0.87 \mu\text{m}$ and $3.0 \times 10^{18} \text{ cm}^{-3}$ doping concentration, the $I_{D\text{MAX}}$ of same doping device with $L_{CH} = 0.3 \mu\text{m}$ increased from 600 to 750 mA/mm for D-mode. D-mode GOOI FET with n of $3 \times 10^{18} \text{ cm}^{-3}$ and L_{CH} of 0.3 μm has a R_C , R_{SH} , and R_{on} of 4.5 $\Omega \text{ mm}$, 5.6 k Ω/\square , and 11 $\Omega \text{ mm}$, showing that the total contact resistance ($2R_C$) of 9 $\Omega \text{ mm}$ is dominate. Therefore, further scaling the L_{CH} shows no significant effect in boosting the $I_{D\text{MAX}}$, which is mainly limited by the high R_C of lower doped channel underneath the contacts. The much improved $I_{D\text{MAX}}$ of 1.5 A/mm mainly originates from the higher doping concentration induced lower R_C rather than the simply channel length scaling.²⁶ The high doping device also has a much lower R_{on} of 5.3 $\Omega \text{ mm}$ compared with that of 11 $\Omega \text{ mm}$ with $3.0 \times 10^{18} \text{ cm}^{-3}$ doping concentration and this significant

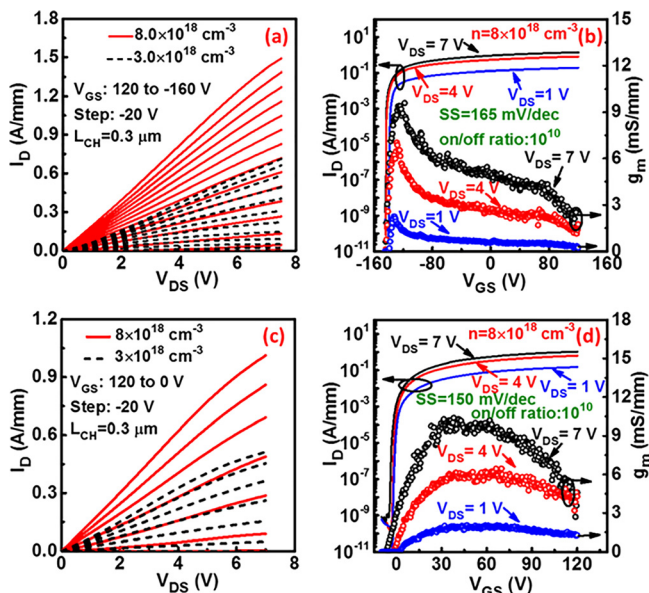


FIG. 2. (a) and (c) I_D - V_{DS} output characteristics of D-mode and E-mode GOOI FETs with 3.0×10^{18} and $8.0 \times 10^{18} \text{ cm}^{-3}$ doping channel, respectively. (b) and (d) I_D - g_m - V_{GS} transfer characteristics of D-mode and E-mode GOOI FETs with $8.0 \times 10^{18} \text{ cm}^{-3}$ doping channel, respectively. Record high $I_{D\text{MAX}}$ of 1.5 and 1.0 A/mm are demonstrated for D/E mode devices. Both D and E-mode devices have high on/off ratio of 10^{10} and low SS of 150–165 mV/dec for 300 nm SiO₂.

lower R_{on} is mostly from the much reduced R_C of 1.7 $\Omega \text{ mm}$. At low V_{DS} regime, high doping I_D - V_{DS} shows linear behavior with lower R_C while the lower doping counterpart displays a Schottky-like contacts with higher R_C . Figure 2(b) is the log-scale I_D - g_m - V_{GS} transfer characteristics of the same D-mode GOOI FET with $8.0 \times 10^{18} \text{ cm}^{-3}$ doping concentration. This D-mode GOOI FET has a threshold voltage (V_T) of -135 V , extracted from the log-scale I_D - V_{GS} at $V_{DS} = 1 \text{ V}$ and $I_D = 0.1 \text{ mA/mm}$. A peak transconductance (g_{max}) of 9.2 mS/mm is achieved which is 2 times of the g_{max} with lower doping concentration, showing the much improved R_C of the higher doping concentration device. Figures 2(c) and 2(d) depict the I_D - V_{DS} output and I_D - g_m - V_{GS} transfer characteristics of an E-mode GOOI FET with $L_{CH} = 0.3 \mu\text{m}$, $t = 55 \text{ nm}$, and $W = 0.17 \mu\text{m}$ for $8.0 \times 10^{18} \text{ cm}^{-3}$, and $t = 75 \text{ nm}$ and $W = 0.45 \mu\text{m}$ for $3.0 \times 10^{18} \text{ cm}^{-3}$ device also shown in Fig. 2(c) as black dashed curves for comparison. Similar to D-mode devices, lower doped E-mode GOOI FETs have an increased $I_{d\text{max}}$ from 450 mA/mm to 550 mA/mm when the L_{CH} is scaled from 1.3 μm to 0.3 μm . Higher E-mode $I_{D\text{MAX}}$ is also achieved with higher doping concentration induced lower R_C of 0.75 $\Omega \text{ mm}$ compared to that of 1.2 $\Omega \text{ mm}$ with lower doping concentration. A record high $I_{D\text{MAX}} = 1.0 \text{ A/mm}$ for higher doping channel is obtained, which is more than 80% higher than lower doping channel. E-mode GOOI FET has a V_T of 2 V determined from the I_D - V_{GS} at $V_{DS} = 1 \text{ V}$ and $I_D = 0.1 \text{ mA/mm}$. Unfortunately, no I_D saturation is observed since applying higher V_{DS} will lead to an abrupt I_D increase and then device breakdown. The drain induced barrier lowering (DIBL) is extracted to be 0.73 and 0.38 V/V for D/E-modes devices, respectively. Finally, benefited from its wide-bandgap and high quality interface between β -Ga₂O₃ and SiO₂, both D/E-mode devices have achieved high on/off ratio of 10^{10} and low subthreshold slope (SS) of 150–165 mV/dec for 300 nm SiO₂.

The significant V_T shift with respect to different β -Ga₂O₃ nano-membrane thickness is due to the surface depletion effect of the unpassivated GOOI FET surface. This is because Sn-doped β -Ga₂O₃ is a 3D semiconductor, which has dangling bonds and surface states on the device surface. Surface depletion could deplete the whole β -Ga₂O₃ nano-membrane tens of nanometers thick. This is the reason why E-mode GOOI FETs can also be realized in high doping β -Ga₂O₃ nano-membrane. The surface depletion effect is verified by using atomic layer deposition (ALD) to deposit 15 nm Al₂O₃ on top to passivate the top surface. As shown in Fig. 1, the V_T is significantly shifted to the left for more than 70 V after the ALD passivation, showing the existence of top and bottom surface depletion on unpassivated GOOI FET surfaces. Similar surface or interface charges induced depletion effect was also observed by Moser *et al.*⁷ Based on the surface depletion, we can obtain each surface depleted charge density (n_s) by using the technology computer aided design (TCAD) C-V simulation to match the measured and simulated V_T from E-mode devices with V_T near zero. The n_s is determined and simulated to be $1.2 \times 10^{13} \text{ cm}^{-2}$ and $2.2 \times 10^{13} \text{ cm}^{-2}$ for $3.0 \times 10^{18} \text{ cm}^{-3}$ and $8.0 \times 10^{18} \text{ cm}^{-3}$ nano-membranes with thickness of 80 nm and 55 nm, respectively. Therefore, the flat-band voltage (V_{FB}) for lower doped and high doped devices are determined to be 135 V and 235 V through the equation

$V_{FB} = \Phi_{MS}/e - 2n_s/C_{ox}$, where Φ_{MS} , e , and C_{ox} are gate-semiconductor work function difference, electron charge quantity, and oxide capacitance, respectively. Higher n_s for higher doping β -Ga₂O₃ nano-membrane is most likely related to the higher surface states with more Sn⁴⁺ dopants. Therefore, the actual C-V curve and I_D - V_{GS} curve are significantly shifted to the right compared with the ideal case without considering surface depletion. Figure 2 shows the simulated C-V curve for E-mode GOOI FET and the V_T from C-V simulation is in good agreement with the V_T from I_D - V_{GS} characterization. Figure 3 is the simulated band diagram of the E-mode GOOI FET at $V_{GS} = 0$ V with lower doping and high doping channels by considering surface depletion effect. The top and bottom depletion regions pull up the conduction band of β -Ga₂O₃ with very few carriers left behind in the nano-membrane.

As a low thermal conductivity material and also its substrate, the heat dissipation is a big issue for β -Ga₂O₃ devices which needs to be seriously considered. We have used an ultra-fast TR set-up to study its thermal property. The system has a high-speed LED pulse illumination and a CCD camera to image temperature dependent reflectance change. Briefly, the source/drain Au pads are illuminated through an LED ($\lambda = 530$ nm), and the change in reflectance under bias is calibrated with Au thermo-reflectance coefficient to translate into temperature change. Figure 4(a) is the CCD camera and TR image merged view of another D-mode GOOI FET device with $V_{DS} = 4$ V and $V_{GS} = 0$ V with $I_D = 0.3$ A/mm. Figure 4(b) shows the 3D view of the TR image along the channel length and channel width directions. Even at a low bias power regime ($P = V_{DS} \times I_D = 1.2$ W/mm), the local temperature has increased by 35 °C compared to room temperature or unbiased devices. It seems that there is a ‘‘cold’’ channel between source and drain contact. This is because

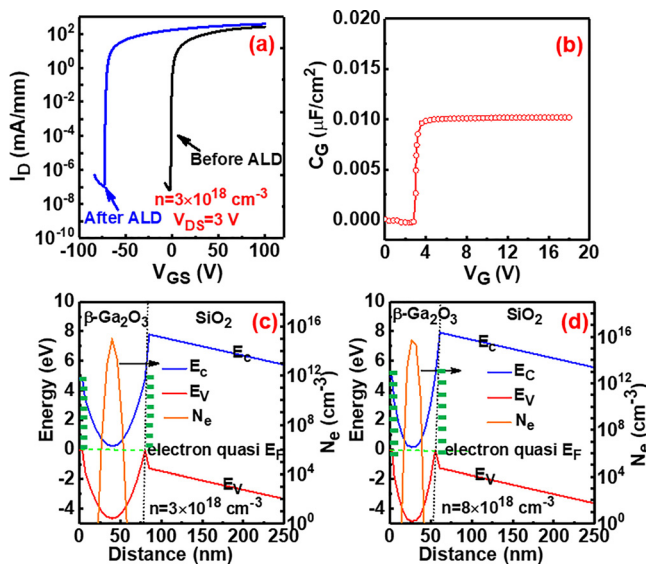


FIG. 3. (a) I_D - V_{GS} comparison between GOOI FETs with and without ALD passivation for β -Ga₂O₃ nano-membrane with doping concentration of 3.0×10^{18} cm⁻³, (b) simulated C-V curve for E-mode GOOI FET at a β -Ga₂O₃ nano-membrane thickness of 80 nm and doping concentration of 3.0×10^{18} cm⁻³ after considering the top and bottom negative surface charge ($n_s = 1.2 \times 10^{13}$ cm⁻²) depletion effect. Band diagram and electron density distribution of E-mode GOOI FETs with surface negative charge depletion on (c) lower doping ($n_s = 1.2 \times 10^{13}$ cm⁻²) and (d) high doping ($n_s = 2.2 \times 10^{13}$ cm⁻²) β -Ga₂O₃ nano-membrane channels at $V_{GS} = 0$ V.

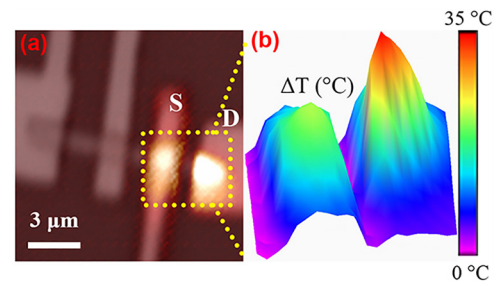


FIG. 4. (a) CCD camera and TR image merged temperature change on the source and drain sides of back-gate GOOI FETs. (b) 3D view of the TR image of GOOI FET. The device is biased at $V_{DS} = 4$ V, $V_{GS} = 0$ V, and $I_D = 0.3$ A/mm. The temperature is calibrated for the S and D gold electrodes.

the thermal reflectance coefficient of β -Ga₂O₃ channel is more than 10 times lower than that of Au electrodes. The temperature measurement is only calibrated with Au surface and we use Au electrodes as the thermometer of the device. Higher power bias will increase device temperature more significantly, and lead to degrade electron mobility and reliability, and eventually breakdown the device. Other self-heating effects induced I_D reduction was observed by Moser and Wong *et al.*^{27,28} The work by applying large thermal conductivity substrates and advanced device structures to mitigate the self-heating effect for β -Ga₂O₃ FETs is on-going and will be reported elsewhere.

We have demonstrated record high $I_{D,MAX}$ of 1.5/1.0 A/mm for D/E-mode GOOI FETs by increasing the β -Ga₂O₃ doping concentration from 3.0×10^{18} to 8.0×10^{18} cm⁻³ and further scaling the channel length. The significant V_T shift with respect to different β -Ga₂O₃ nano-membrane thicknesses is due to the surface depletion effect of the unpassivated GOOI FET surface. High on/off ratio of 10^{10} and low SS of 150 mV/dec are achieved. Self-heating effect is also directly observed with the TR measurement. GOOI FETs with wide bandgap, high $I_{D,MAX}$, and low R_c offer the promise in the power device applications if the low thermal conductivity issue can be solved.

The authors thank the technical guidance from the Sensors Directorate of Air Force Research Laboratory.

- ¹M. Higashiwaki, K. Sasaki, H. Murakami, Y. Kumagai, A. Koukitsu, A. Kuramata, T. Masui, and S. Yamakoshi, *Semicond. Sci. Technol.* **31**, 034001 (2016).
- ²M. H. Wong, K. Sasaki, A. Kuramata, S. Tamakoshi, and M. Higashiwaki, *IEEE Electron Device Lett.* **37**, 212 (2016).
- ³A. J. Green, K. D. Chabak, E. R. Heller, R. C. Fitch, M. Baldini, A. Fiedler, K. Irmischer, G. Wagner, Z. Galazka, S. E. Tetlak, A. Crespo, K. Leedy, and G. H. Jessen, *IEEE Electron Device Lett.* **37**, 902 (2016).
- ⁴K. D. Chabak, N. Moser, A. J. Green, D. E. Walker, Jr., S. E. Tetlak, E. Heller, A. Crespo, R. Fitch, J. P. McCandless, K. Leedy, M. Baldini, G. Wagner, Z. Galazka, X. Li, and G. Jessen, *Appl. Phys. Lett.* **109**, 213501 (2016).
- ⁵M. Higashiwaki, K. Sasaki, A. Kuramata, T. Masui, and S. Yamakoshi, *Appl. Phys. Lett.* **100**, 013504 (2012).
- ⁶K. Konishi, K. Goto, H. Murakami, Y. Kumagai, A. Kuramata, S. Yamakoshi, and M. Higashiwaki, *Appl. Phys. Lett.* **110**(10), 103506 (2017).
- ⁷N. A. Moser, J. P. McCandless, A. Crespo, K. D. Leedy, A. J. Green, E. R. Heller, K. D. Chabak, N. Peixoto, and G. H. Jessen, *Appl. Phys. Lett.* **110**, 143505 (2017).

- ⁸A. J. Green, K. D. Chabak, M. Baldini, N. Moser, R. C. Gilbert, R. Fitch, G. Wagner, Z. Galazka, J. Mccandless, A. Crespo, K. Leedy, and G. H. Jessen, *IEEE Electron Device Lett.* **38**, 790 (2017).
- ⁹K. Imscher, Z. Galazka, M. Pietsch, R. Uecker, and R. Fornari, *J. Appl. Phys.* **110**, 063720 (2011).
- ¹⁰Z. Galazka, K. Imscher, R. Uecker, R. Bertram, M. Pietsch, A. Kwasniewski, M. Naumann, T. Schulz, R. Schewski, D. Klimm, and M. Bickermann, *J. Cryst. Growth* **404**, 184 (2014).
- ¹¹H. Aida, K. Nishighuchi, H. Takeda, N. Aota, K. Sunakawa, and Y. Yaguchi, *Jpn. J. Appl. Phys., Part 1* **47**, 8506 (2008).
- ¹²A. Kuramata, K. Koshi, S. Watanabe, Y. Yamaoka, T. Masui, and S. Yamakoshi, *Jpn. J. Appl. Phys., Part 1* **55**, 1202A2 (2016).
- ¹³N. Ueda, H. Hosono, R. Waseda, and H. Kawazoe, *Appl. Phys. Lett.* **70**, 3561 (1997).
- ¹⁴E. G. Villora, K. Shimamura, Y. Yoshikawa, K. Aoki, and N. Ichinose, *J. Cryst. Growth* **270**, 420 (2004).
- ¹⁵M. D. Santia, N. Tandon, and J. D. Albrecht, *Appl. Phys. Lett.* **107**, 041907 (2015).
- ¹⁶Z. Guo, A. Verma, X. Wu, F. Sun, A. Hickman, T. Masui, A. Kuramata, M. Higashiwaki, D. Jena, and T. Luo, *Appl. Phys. Lett.* **106**, 111909 (2015).
- ¹⁷H. Zhou, M. Si, S. Alghamdi, G. Qiu, L. Yang, and P. D. Ye, *IEEE Electron Device Lett.* **38**, 103 (2017).
- ¹⁸W. S. Hwang, A. Verma, H. Peelaers, V. Protasenko, S. Rouvimov, H. Xing, A. Seabaugh, W. Haensch, C. Van de Walle, Z. Galazka, M. Albrecht, R. Fornari, and D. Jena, *Appl. Phys. Lett.* **104**, 203111 (2014).
- ¹⁹S. Krishnamoorthy, Z. Xia, S. Bajaj, M. Brenner, and S. Rajan, *Appl. Phys. Express* **10**, 051102 (2017).
- ²⁰M. H. Wong, Y. Nakata, A. Kuramata, S. Yamakoshi, and M. Higashiwaki, *Appl. Phys. Express* **10**, 041101 (2017).
- ²¹K. Maize, J. Christofferson, and A. Shakouri, in *Proceedings of the 24th IEEE SEMI-THERM Symposium* (2008), p. 55.
- ²²H. Zhou, S. Alghamdi, M. W. Si, and P. D. Ye, *IEEE Electron Device Lett.* **37**, 1411 (2016).
- ²³K. Maize, A. Ziabari, W. D. French, P. Lindorfer, B. OConnell, and A. Shakouri, *IEEE Trans. Electron Devices* **61**, 3047–3053 (2014).
- ²⁴S. H. Shin, M. A. Wahab, M. Masduzzaman, K. Maize, J. J. Gu, M. Si, A. Shakouri, P. D. Ye, and M. A. Alam, *IEEE Trans. Electron Devices* **62**, 3516 (2015).
- ²⁵M. Farzaneh, K. Maize, D. Lüerßen, J. A. Summers, P. M. Mayer, P. E. Raad, K. P. Pipe, A. Shakouri, R. J. Ram, and J. A. Hudgings, *J. Phys. D: Appl. Phys.* **42**, 143001 (2009).
- ²⁶H. Zhou, Y. Du, and P. D. Ye, *Appl. Phys. Lett.* **108**, 202102 (2016).
- ²⁷N. A. Moser, A. Crespo, S. E. Tetlak, A. J. Green, K. D. Chabak, and G. H. Jessen, in *74th IEEE Device Research Conference* (Technical Digest, 2016), p. 95.
- ²⁸M. H. Wong, Y. Morikawa, K. Sasaki, A. Kuramata, S. Yamakoshi, and M. Higashiwaki, *Appl. Phys. Lett.* **109**, 193503 (2016).